# **Pre-mixed Partially Alloyed Iron Powder for Warm Compaction: KIP Clean Mix HW Series**\*





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powder metallurg . Material C containing no graphite was prepared to investigate the powder behavior during warm compaction. Material D was solel composed of base iron powder and was prepared for comparison with material C.

### **2.1.2** Compacting and sintering conditions and sinterior sint

Compacting and sintering conditions for materials A and B are as shown is **2**. (For convenience sake, warm compaction and cold compaction are hereafter abbreviated as WC and CC respectivel in the tables and figures.)

## $2.1.3 -$

Sintered materials obtained under the conditions listed in Table 2 were subjected to carburi ation for 9 ks at 1 193 K and carbon potential of 0.9 mass%, and then for 2.64 ks at 1 123 K and carbon potential of

of particle rearrangement and those of plastic deformation, respectivel. The value on the left-hand side of Eq. (3),  $C_{\varepsilon}(p)$ , was called the porosit reduction ratio. The constants in Eq. (2), *a*, *b*, *A*, and *B* were abtained from measured values b means of the non-linear leastsquares method. The contributions of particle rearrangement and plastic deformation in the compacting process were estimated b comparing the magnitudes of the constants, *a* and *A*.

#### **2.2.3 Green compact strength**

Green compacts compacted into blocks of 10 mm in width 35 mm in length 3 mm in thickness were subjected to 3-point bending tests to evaluate their transvers rupture strength.

**2.2.4 f and heat-treated compacts, and pore** 

Tensile strength, elongation, impact value, and Rockwell hardness were measured for each test specimen. In addition, contact fatigue strength was measured for warm-compacted, sintered and heat-treated compacts and DPDS-treated compacts. The contact fatigue

apparent densit exhibit only small changes with temperature from ambient temperature to 423 K. This behavior apparently results from the fact that KW-wax has a high melting point and is less-sensitive to temperature changes, which provides the advantage that precise temperature control is not required during warm compaction of this material.

The green densit of material A was higher than that of cold-compacted material B. The temperature dependence tended to decrease in a range be ond 373 K. Its high densit at ambient temperature results from particle rearrangement promoted b KW-wax as will be described later. The decrease in the temperature dependence of the densit in the high temperature range is considered as the result that KW-wax does not melt in this temperature region due to its high melting point and partiall remains within the green compact.

## **3.2 Compacting Behavior in Warm Compaction**

**Figure 4** shows the compaction pressure dependence of the reduction rate of porosities of materials C and D at 433 K. The values observed in the experiment and those calculated by the regression equation are in a good agreement. Figure 4 also includes the values obtained for the <sub>f</sub>est term,  $C_{\epsilon_1}(p)$ , and the second term,  $C_{\epsilon_2}(p)$ , in the regression equation. At a constant compaction pressure, material C exhibited a higher porosit reduction ratio than material D which was solely composed of iron powder, demonstrating the advantage provided b the lubricant KW-wax in obtaining a higher-densit compact. Comparing  $C_{\varepsilon,1}(p)$  and  $C_{\varepsilon,2}(p)$  of each material, it is apparent that  $C_{\epsilon,1}(p)$  of material C increases steeply at a relativel low compaction pressure and then approaches a constant value. This behavior occurs because the lubricant KW-wax added to material C promotes the particle rearrangement even at a low compaction pressure and contributes to the densi cation of the powder. It was previously reported that the densi cation of material b

warm compaction is attributable to an increase of plastic deformation capabilit of iron powder b heating.<sup>9)</sup> Thus, HW Series iron powder pre-mix with KW-wax 49.95ses 3Tc



Photo 2 Microstructure of the center part of as-sintered material A

**3.4 Mechanical Properties of Sintered Material**

**3** and **4** show sintered densit, dimensional change, Charpy impact value, hardness, tensile strength, elongation, and contact fatigue strength of sintered material and heat-treated material respectivel .

**and Heat-treated Material**

The tensile strength, impact value and hardness increase in proportion to the increase in densit, thus con <sub>f</sub>rming the effect of densi cation by warm compaction. **1** and **2**, respectivel, show the microstructures near the surface and in the center part of warm-compacted and sintered compact, having a densit of  $7.38$  Mg/m<sup>3</sup>

results of anal ing the compacting behavior. The mechanical properties of warm-compacted and sintered material are thus improved not only by the increase in densit but also by promotion of particle rearrangement, which leads to the uniform distribution of nearly circular pores of si e with narrow distribution throughout the sintered compact.

b a process of warm-compacting, sintering and carburi ing heat-treatment, tends to decrease with increasing densit. This behavior is interpreted that the decrease in ratio of open pores in the surface area with increase in densit, suppresses carburi ation inside materials. In the gas-carburi ing treatment of warm-compacted and sintered material, a future issue is to optimi e the carburi ing conditions.

#### **References**

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